*Research Article*

# Investigation into the Influence of Substitutions on the Structural Characteristics of $\text{Sr}_2\text{FeMoO}_6$

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Received: 05.11.2025      Accepted: 09.11.2025      Published: 19.11.2025

<https://doi.org/10.54414/XVZB1367>

## Abstract

$\text{Sr}_2\text{FeMoO}_6$  is a half-metallic ferromagnet with a high Curie temperature, which positions it as a strong candidate for spintronic devices. Its lattice features ordered Fe and Mo cations on the B sites, and this arrangement underpins distinctive electronic transport and robust magnetic exchange. Here we examine  $\text{Sr}_2\text{FeMoO}_6$  and the substituted phases  $\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$  with  $x = 0.05$  and  $\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$  with  $x = 0.15$  by neutron diffraction across multiple temperatures. The data show that temperature and B-site substitution drive measurable changes in unit-cell metrics and anisotropy. These results clarify the structural and thermal responses of this double perovskite family and inform their use in technology.

**Keywords:** neutron diffraction, rietveld refinement, substitutions, low temperature

## 1. Introduction

$\text{Sr}_2\text{FeMoO}_6$  is a double perovskite oxide that exhibits half-metallic ferromagnetism together with a high Curie temperature, which makes it a strong candidate for spintronic devices. Its lattice hosts an ordered arrangement of Fe and Mo cations on the B sites, and this chemical ordering drives unusual electronic transport and robust magnetic exchange. As a result, the material combines near-perfect spin polarization with stable long-range magnetism at technologically relevant temperatures [1]. Furthermore,  $\text{Sr}_2\text{FeMoO}_6$  has been explored for applications in solid oxide fuel cells due to its strong electronic conductivity and thermal resilience [2].

$\text{Sr}_2\text{FeMoO}_6$  is a B-site-ordered double perovskite that combines highly spin-polarized conduction with stable ferrimagnetic behavior at and above room temperature. Its magnetic and electronic performance originates from the distinctive interaction between Fe and Mo ions and the resulting band structure [3]. Achieving high magnetization, strong spin polarization, and reliable transport properties depends on strict control of composition, oxygen non-stoichiometry, and cation ordering in both bulk samples and thin film architectures [4]. Because its defect chemistry can be adjusted over a wide range,  $\text{Sr}_2\text{FeMoO}_6$  serves as a flexible material platform for spintronic elements and magnetoresistive devices used in oxide-based electronics [5]. Tailored microstructures and composite configurations also produce strong responses in the radio frequency and microwave regimes, which makes the material useful for sensing technologies and electromagnetic interference management [6]. In addition to these functions, the combined electrical conductivity, redox durability, and ability to accommodate dopants make  $\text{Sr}_2\text{FeMoO}_6$  a promising option for electrocatalytic applications and as an anode material in solid oxide electrochemical systems [1].

$\text{Sr}_2\text{FeMoO}_6$  is a B-site ordered double perovskite that shows half-metallicity and robust ferrimagnetism near room temperature. Large low-field magnetoresistance arises from spin-polarized tunneling across grain

boundaries. This result established SFMO as a model oxide for spintronics [7]. Neutron diffraction and magnetometry confirm ferrimagnetic ordering arising from Fe<sup>3+</sup>–O–Mo<sup>5+</sup> pathways in ordered SFMO and reveal how even modest Fe/Mo antisite disorder measurably depresses the refined magnetic moments and transition temperature [8]. Room-temperature neutron data on SFMO– $\delta$  ceramics underscore that oxygen-defect chemistry subtly distorts the cell and modulates ordered moments, reinforcing the tight coupling among oxygen content, structural metrics, and ferri-magnetic order [9].

The study [10] revealed that increasing the total sintering time significantly improved Fe/Mo cation ordering in Sr<sub>2</sub>FeMoO<sub>6</sub>, reducing antisite defects and increasing the degree of ordering from 44.9% to 80.2%. Rietveld refinement showed a gradual decrease in unit cell volume and bond lengths (Fe–O and Mo–O) with extended sintering, indicating a more compact and ordered crystal structure. Additionally, the Fe–O1–Mo bond angle increased with sintering time, reflecting reduced octahedral tilting and enhanced linearity along the Fe–O–Mo chains. A structural phase transition from cubic (*Fm $\bar{3}m$* ) to tetragonal (*I4/m*) symmetry occurs in Sr<sub>2</sub>FeMoO<sub>6</sub> near 400 K, accompanied by octahedral rotations and the onset of ferrimagnetic ordering. Neutron diffraction and Mössbauer spectroscopy revealed a saturated Fe magnetic moment of  $\sim 4.3$   $\mu_B$  and an intermediate Fe valence state, with approximately 6% Mo-site vacancies contributing to local structural and magnetic inhomogeneity [11]. Cation ordering in the *I4/m* structure and antisite defects control the balance between high saturation magnetization, metallic transport, and spin polarization. Targeted A- or B-site substitutions and careful oxygen control tune the Curie temperature and magnetoresistance. This review frames the structure–property rules that guide later work [12]. Systematic neutron refinements of Sr<sub>2</sub>Fe<sub>0.75</sub>T<sub>0.25</sub>MoO<sub>6</sub> (T = Cr, Mn, Co, Ni) demonstrate dopant-specific lattice distortions and disorder that degrade  $M_s$  and metallicity to differing degrees, mapping how 3d-dopant chemistry perturbs Fe–O–Mo exchange [13]. In Sr<sub>2</sub>Fe<sub>1-x</sub>Cr<sub>x</sub>MoO<sub>6-y</sub>, increasing Cr systematically reduces  $M_s$  and increases resistivity by disrupting Fe–O–Mo double-exchange and promoting antisites/oxygen defects, providing a clear, composition-dependent picture of structure–magnetism coupling [14].

Magnetocaloric measurements on SFMO and W-doped analogs link structural distortions to the magnetic-entropy change, indicating that judicious B-site chemistry can shift operating windows toward technologically relevant temperatures [15].

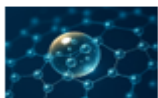
In this context, our research focuses on further understanding the structural behavior of Sr<sub>2</sub>FeMoO<sub>6</sub>-based systems under different doping strategies and temperature conditions using high-resolution neutron diffraction and Rietveld analysis.

## 2. Research Method

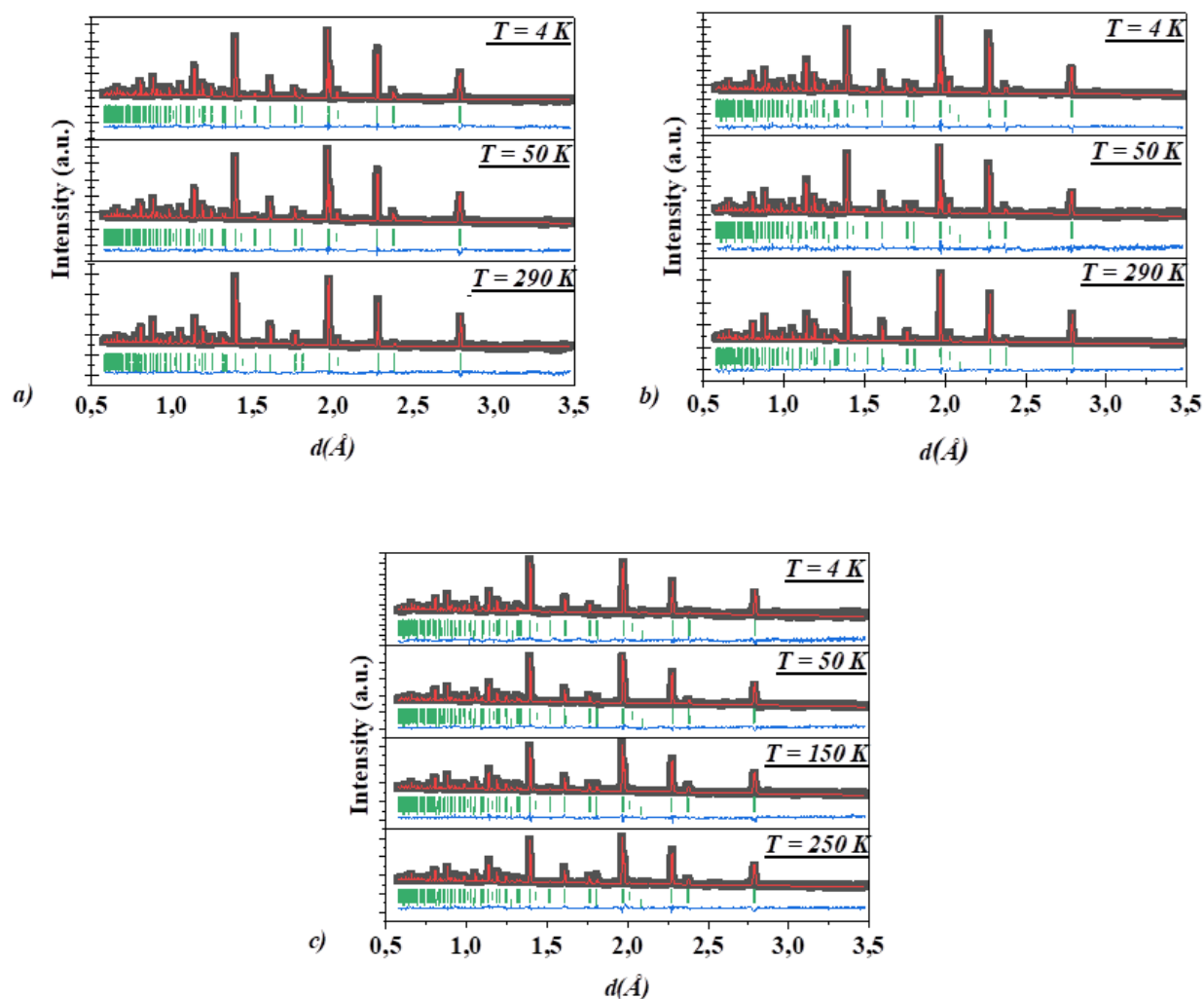
Neutron diffraction measurements were conducted over a range of temperatures using the High-Resolution Fourier Diffractometer (HRFD) at the Frank Laboratory of Neutron Physics, Joint Institute for Nuclear Research (JINR), Dubna, Russia. The instrument provides high angular resolution and is optimized for the precise determination of lattice parameters and atomic positions, allowing accurate tracking of temperature-induced structural variations in the studied samples [16]. The purpose was to investigate the structural behavior of the parent compound Sr<sub>2</sub>FeMoO<sub>6</sub> and its cation-substituted derivatives, namely Sr<sub>2</sub>FeMo<sub>1-x</sub>Ta<sub>x</sub>O<sub>6</sub> ( $x = 0.05$ ) and Sr<sub>2</sub>FeMo<sub>1-x</sub>V<sub>x</sub>O<sub>6</sub> ( $x = 0.15$ ). Samples were synthesized with controlled stoichiometry and uniform microstructure to ensure reliable diffraction data. Temperature and B-site substitution with Ta and V were varied systematically to assess their impact on unit-cell metrics. Lattice parameters were obtained by full-profile Rietveld refinement, which also yielded fit statistics and peak-broadening indicators. These measurements provide a consistent basis for interpreting thermal and substitution effects in this double perovskite system [17,18].

## 3. Results and Discussion

Neutron diffraction experiments were conducted on Sr<sub>2</sub>FeMoO<sub>6</sub>-based samples at varying dopant concentrations and temperatures. The diffraction spectra were refined using the Rietveld method implemented in the FullProf Suite software [19,20].



In the initial undoped state, the diffraction pattern illustrates a well-defined structure, from which changes in lattice parameters with temperature were identified (Figure 1).



**Figure 1.** Neutron diffraction patterns of (a) initial  $\text{Sr}_2\text{FeMoO}_6$ , (b)  $\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$  ( $x = 0.15$ ), and (c)  $\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$  ( $x = 0.05$ ) at different temperatures. Black dots represent experimental data, the red line is the Rietveld refinement fit, green vertical ticks indicate Bragg reflection positions, and the blue line shows the difference between observed and calculated intensities.

Following V doping at  $x = 0.15$ , diffraction patterns were collected at three temperatures, as shown in Fig. 1b. In addition to the perovskite reflections, weak extra peaks appear that do not index to the  $\text{Sr}_2\text{FeMoO}_6$  cell and are assigned to aluminum and copper from the sample environment used to secure the containers under the measurement conditions. In the Ta doped compound  $\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$  with  $x = 0.05$ , the unit cell volume remains close to that of the undoped sample, with a slight increase at intermediate temperature and reaching  $245.95 \text{ \AA}^3$  at 250 K as shown in Fig. 1c. The a axis shows a minor expansion while the c axis is nearly temperature independent, which indicates that the framework is largely preserved with Ta substitution. The full set of refined structural parameters is summarized in Table 1. For the undoped  $\text{Sr}_2\text{FeMoO}_6$ , the unit cell volume increases with temperature from  $244.63 \text{ \AA}^3$  at 4 K to  $246.14 \text{ \AA}^3$  at 290 K, consistent with moderate thermal expansion.

For the V-doped composition  $\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$  with  $x$  equal to 0.15, the unit cell volume remains smaller at all temperatures when compared with the undoped material and the Ta-substituted sample [21]. At 4 K, the volume reaches its lowest value of  $243.99 \text{ \AA}^3$ . The c-axis shows the strongest decrease, which indicates that the

introduction of vanadium produces a clear lattice contraction. This shrinkage is more evident than the effect produced by Ta substitution and demonstrates the stronger structural impact of vanadium on the  $\text{Sr}_2\text{FeMoO}_6$  lattice.

**Table 1.** The changes in structural parameters after substitution.

Space group I 4/m m m	a	c	V
$\text{Sr}_2\text{FeMoO}_6$ T = 4 K	5.5599(1)	7.9134(1)	244.627(6)
$\text{Sr}_2\text{FeMoO}_6$ T = 50 K	5.5607(1)	7.9138(2)	244.71(1)
$\text{Sr}_2\text{FeMoO}_6$ T = 290 K	5.5785(1)	7.9096(2)	246.14(1)
$\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$ x=0.15, T = 4 K	5.5560(1)	7.9040 (2)	243.994 (1)
$\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$ x=0.15, T = 50 K	5.5547(1)	7.9039(2)	243.98(1)
$\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$ x=0.15, T = 290 K	5.5730(1)	7.8986(2)	245.32(1)
$\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$ x=0.05, T = 4 K	5.5628 (2)	7.9059(2)	244.778(2)
$\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$ x=0.05, T = 50 K	5.5628(2)	7.9101(3)	244.78(1)
$\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$ x=0.05, T = 150 K	5.5676 (2)	7.9085 (3)	245.15(2)
$\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$ x=0.05, T = 250 K	5.576(1)	7.909(1)	245.95(5)

These findings emphasize the role of B-site substitution in tuning the thermal and structural behavior of double perovskites. While Ta maintains structural integrity across temperature ranges, V leads to distinct lattice shrinkage, underlining the importance of dopant selection for potential applications where thermal and structural stability are critical.

#### 4. Conclusion

Taken together, the results show that temperature and B-site substitution strongly govern the crystal structure of  $\text{Sr}_2\text{FeMoO}_6$ . Thermal expansion increases the unit-cell volume of the pristine compound, and the effect is anisotropic along the a and c axes. Ta incorporation keeps the lattice essentially unchanged and points to good structural robustness under heating. V substitution produces a clear lattice contraction, most notably along the c axis, consistent with a modified bonding environment and altered octahedral distortions. These insights identify B-site chemistry as an effective handle to tailor thermal and structural responses in double perovskites, and they guide optimization for spintronic and solid-oxide energy applications.

#### Author Contributions

Dilmurod A. Rakhmanov: conceptualization methodology, writing original draft.

Vugar H. Mirzayev: conceptualization, funding, acquisition, project administration.

#### Funding

No funding was received for this research.

#### Conflicts of Interest

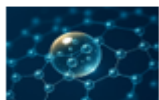
The authors declare that they have no conflicts of interest.

#### Supplementary Information

Not applicable.

#### Ethical Approval

Not applicable.



## Acknowledgments

The authors thank A.S. Abiyev for his assistance in conducting the experiment.

## Data and Code Availability

The data supporting the findings of this study are available from the corresponding author upon reasonable request. No custom code was used in this study.

## Abbreviations

Strontium Iron Molybdate ( $\text{Sr}_2\text{FeMoO}_6$ ), Iron (Fe), Molybdenum (Mo), Strontium Iron Molybdenum Tantalum Oxide ( $\text{Sr}_2\text{FeMo}_{1-x}\text{Ta}_x\text{O}_6$ ), Strontium Iron Molybdenum Vanadium Oxide ( $\text{Sr}_2\text{FeMo}_{1-x}\text{V}_x\text{O}_6$ ), SFMO ( $\text{Sr}_2\text{FeMoO}_6$ ), Chromium (Cr), Manganese (Mn), Cobalt (Co), Nickel (Ni), Saturation Magnetization ( $M_s$ ), High Resolution Fourier Diffractometer (HRFD), Joint Institute for Nuclear Research (JINR).

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